

# Process Intensification by CO<sub>2</sub> for High Quality Carbon Nanotube Forest Growth: Double-Walled Carbon Nanotube Convexity or Single-Walled Carbon Nanotube Bowls?

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Received: 13 July 2009 / Revised: 2 September 2009 / Accepted: 13 September 2009

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## ABSTRACT

Introduction of CO<sub>2</sub> is a facile way to tune the growth of vertically aligned double- or single-walled carbon nanotube (CNT) forests on wafers. In the absence of CO<sub>2</sub>, a double-walled CNT convexity was obtained. With increasing concentration of CO<sub>2</sub>, the morphologies of the forests transformed first into radial blocks, and finally into bowl-shaped forests. Furthermore, the wall number and diameter distribution of the CNTs were also modulated by varying the amount of CO<sub>2</sub>. With increasing CO<sub>2</sub> concentration, CNTs with fewer wall number and smaller diameter were obtained. The addition of CO<sub>2</sub> is speculated to generate water and serve as a weak oxidant for high quality CNT growth. It can tune the growth rate and the morphologies of the forests, prevent the formation of amorphous carbon, and reduce the wall number of the CNTs.

## KEYWORDS

Carbon nanotube forest, carbon dioxide, chemical vapor deposition, self organization, process intensification

## Introduction

Carbon nanotubes (CNTs) have been extensively studied due to their unique mechanical, electronic, and thermal properties [1–3]. CNTs can be considered as a seamless cylinder formed by rolling graphene. The huge aspect ratio helps to generate anisotropic properties along the *c* axis. In vertically aligned form, CNTs have relatively large aspect ratio and good orientation [1, 2], benefiting both the intrinsic performance of CNTs and the facile application of CNT forests [2–6]. The wall number and the diameter

of CNTs have been found to affect the properties of CNTs significantly, leading to different applications [5]. Recently, both single- and double-walled CNTs have attracted increasing attention due to their promising applications in field effect transistors, and field emission and electric double-layer capacitors [2, 7–9].

In order to control the wall number distribution of the synthesized CNTs, precisely controlled catalyst layers with different thickness are commonly employed to modulate the size of catalyst particles and the CNTs [5, 10–14]. It is believed that the wall

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number distribution depends strongly on the nominal thickness of the catalyst film. Generally, CNTs with fewer walls grow on thinner catalyst films. Deposition through nano-templates can also control the catalyst size, leading to the preferential growth of wall-number selected CNTs [15]. Delicate control of the annealing process has also been adopted to tune the wall number distribution of as-grown CNTs [16].

These results are important for the development of a standard process for mass production of aligned single-walled CNT forests. Process intensification, which is widely accepted by chemical engineers, should be considered to improve the efficiency and lower the production cost of this process [17]. The process of mass production of CNTs, especially single-walled CNTs (SWCNTs), can be greatly intensified by the introduction of a gaseous oxidizer. For example, in the so-called “super growth” of single-walled CNT forests, the product was strongly affected by the water content in the reactor. Oxygen-containing gases have been tested in the chemical vapor deposition (CVD) process, and they showed great ability to modulate the structure of CNTs [18–21]. It is commonly believed that the addition of water or steam can promote the synthesis of CNT forests, prolong the lifetime of catalyst particles, and prevent the sintering of iron particles [22–25]. Water can also open the end of CNTs by selective etching, which also shows great potential for the delicate tuning of the fine structure of CNTs [26, 27]. Furthermore, it should be noted that CO<sub>2</sub> is also a good weak oxidant which promotes the growth of CNTs [28, 29]. Compared with water, the flow rate of CO<sub>2</sub> can be more easily controlled and it does not condense in the pipeline at room temperature. However, the effect of CO<sub>2</sub> promotion on the growth of CNT forests is not yet fully understood [30, 31].

In this contribution, CO<sub>2</sub> was selected as a promoter to develop a highly efficient process for the synthesis of aligned CNTs with controlled morphology and structure. The morphologies of CNT forests were modulated by controlling the concentration of CO<sub>2</sub> in the CVD growth period. By varying the amount of CO<sub>2</sub>, various kinds of forests can be obtained on the same substrate and multi-layered forests can also be fabricated.

## 1. Experimental

Electron beam evaporation was employed to deposit Al<sub>2</sub>O<sub>3</sub> (10.0 nm), Fe (1.0 nm), and MgO (1.0 nm) sequentially on a silicon wafer with SiO<sub>2</sub> of 600 nm thickness as catalyst. The silicon wafer was then cut into small pieces (5.0 mm × 5.0 mm) and put into a quartz tube with an inner diameter of 30 mm and length of 1200 mm for CVD growth. In a typical growth process, the quartz tube containing the substrate was positioned in the tube furnace under the protection of Ar and H<sub>2</sub> and the temperature was raised. When the temperature reached 750 °C, CO<sub>2</sub> and C<sub>2</sub>H<sub>4</sub> were introduced into the quartz tube to start the synthesis of CNT forests. The flow rates of Ar, H<sub>2</sub>, and C<sub>2</sub>H<sub>4</sub> were 250, 200, and 100 standard cubic centimeters per minute (sccm), respectively. The concentration of CO<sub>2</sub> was varied in order to modulate the growth of CNT forests. After a 1-h growth of CNT forests, the supply of C<sub>2</sub>H<sub>4</sub> and CO<sub>2</sub> was shut off and the reaction chamber was cooled down under the protection of Ar and H<sub>2</sub>.

The morphology of the CNT forests was characterized by a JSM 7401F high-resolution scanning electron microscope (SEM) operated at 5.0 kV and a JEM 2010 high-resolution transmission electron microscope (TEM) operated at 120.0 kV. The sample for the TEM observation was prepared by a common sonication method. Raman experiments were performed with a Renishaw RM2000 Raman spectrophotometer. The spectra were recorded using a He–Ne laser excitation line at 633.0 nm with a spot size of about 20 μm<sup>2</sup> on the CNT forest kept at ambient temperature. To investigate the thermal stability of the CNT forests, thermogravimetry (TG) was carried out on a TGA Q500 instrument, using CO<sub>2</sub> as the atmosphere and with a temperature ramp rate of 10 °C/min.

## 2. Results and discussion

### 2.1 Tailoring the morphologies of CNT forests

Figures 1(a)–1(f) show SEM images of the CNT forests grown on a silicon wafer with different CO<sub>2</sub> concentrations. The heights of the CNT forests at

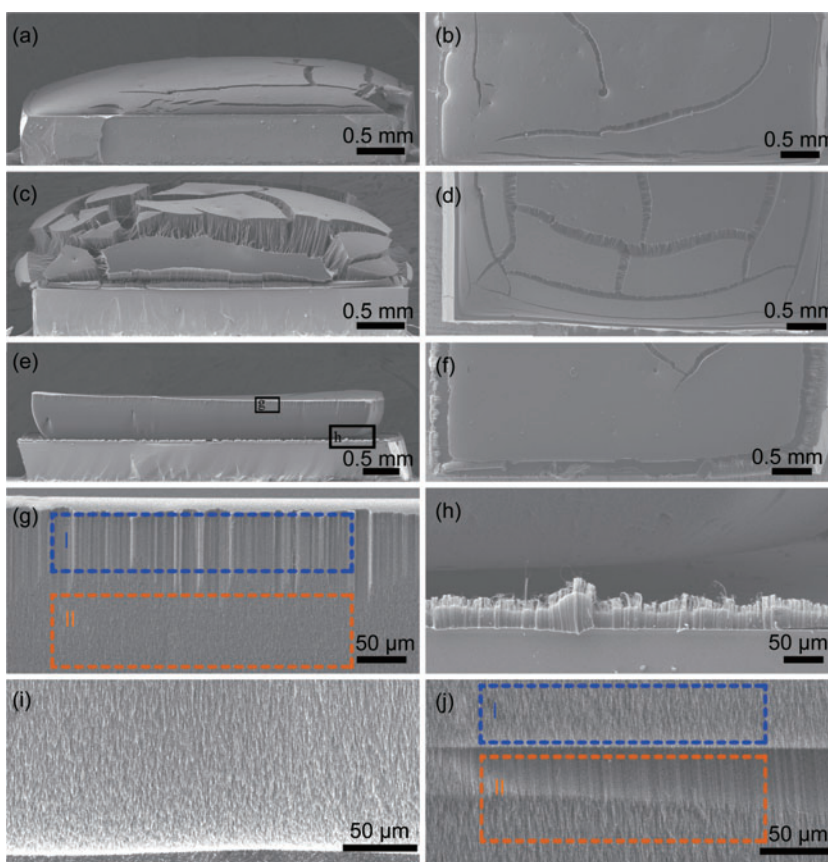


the edges were shorter than those in the center of the substrate (Figs. 1(a), 1(c), and 1(e)). A woven-structured top surface composed of CNTs entangling with each other was formed due to the synchronous growth mode of CNT forests [32]. Vertically aligned convex-shaped, radial-shaped, and bowl-shaped CNT forests can be obtained under different CO<sub>2</sub> concentrations.

A convex-shaped CNT forest was obtained without the addition of CO<sub>2</sub>. The height of the middle part of the forest was about 0.4 mm (Fig. 1(a)). A few cracks (three in this case) can be found on the top of the forest (Fig. 1(b)). The width of the cracks was less than 100 μm. The as-grown CNT forest with a CO<sub>2</sub> concentration of 6.5 mol% is shown in Fig. 1(c). The height of the CNT forests in their middle part reached about 1.1 mm. Compared with the growth without CO<sub>2</sub>, the nonuniformity of the growth rate was larger and the CNT film split into many blocks. As can be seen from Fig. 1(d), over ten cracks were present on the top surface of the forest, and the width of cracks reached 300 μm. The shape of the as-grown CNT forest was similar to that obtained through radial growth on spherical substrates [33]. Due to the difference in growth rate, every block tended to lean towards the edge of the substrate.

Under high CO<sub>2</sub> concentration (30.4 mol%), forests with an arc-shaped bottom were formed by the sequential fracture of CNTs. The as-grown CNT forest split at the edge of the substrate (Fig. 1(e)) and a uniform height of about 0.55 mm was obtained over the whole as-grown forest. The top view of the forest (Fig. 1(f)) showed a continuous top structure with few cracks. The enlarged image of the top part of the forest (Fig. 1(g)) shows the detailed structure where the split occurred. Part I is composed of continuous vertically aligned CNTs and the lines in the image represent

the sides of the assembled CNTs. Part II is composed of the roots of the aligned CNTs, which sequentially rupture at the root from the outermost parts of the whole forest. Due to the continuous growth of the CNT forest and the fracture at the edge of the substrate, CNT roots can be observed from the side of the forest. Figure 1(h) shows the relationship between the forest and the substrate in Fig. 1(e). Some short CNTs are attached to the edge of the substrate, corresponding to the top parts of the CNT forests (Part I in Fig. 1(g)). The rupture of CNT roots from the substrate continued when prolonging the growth duration, and the forest may even detach itself from the substrate over long growth periods under high CO<sub>2</sub> concentrations. Figure 1(i) shows the



**Figure 1** Side view and top view SEM images of CNT forests obtained with CO<sub>2</sub> concentrations of ((a), (b)) 0 mol%, ((c), (d)) 6.5 mol%, and ((e), (f)) 30.4 mol%; (g) enlarged view of the side of CNT forests composed of (I) vertically aligned CNTs and (II) aligned roots of CNTs; (h) enlarged view showing the forest bottom, and the small fraction of CNT forest on the edge and the silicon wafer; (i) the bottom part of the forest composed of aligned CNT roots; (j) the interface of a two-layered CNT forest obtained with 30.4 mol% CO<sub>2</sub>: both layers exhibit bowl-like morphologies with (I) the bottom part of the first layer composed of CNT roots, and (II) the top part of the second layer composed of short vertically aligned CNTs and the aligned CNT roots

bottom of the detached forest, which consists of CNT roots and shows an arc-shaped outer surface. When a two-layered CNT forest was obtained by the carbon source interruption method, the same bowl-like shape was found in both layers of forests (Fig. 1(j)). Part I is the bottom area of the first layer composed of fractured CNTs. Part II shows the second layer composed of an ordered forest structure and the fractured roots, which is similar to that in Fig. 1(g).

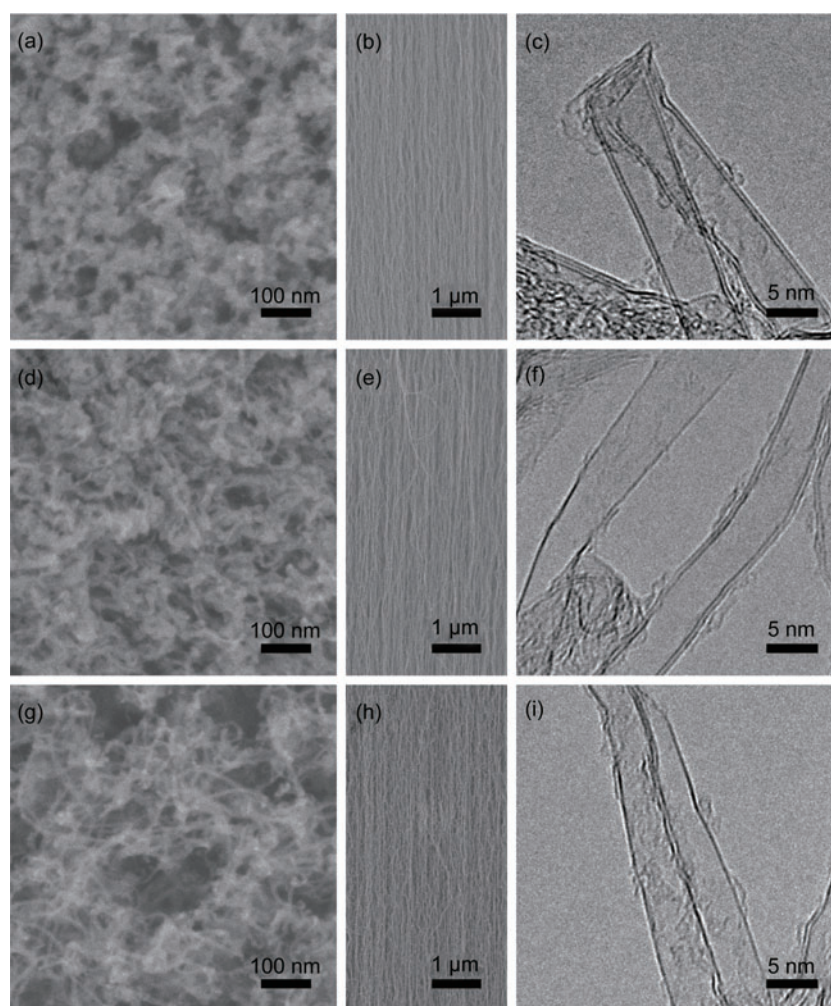
To investigate the generality of this phenomenon, the growth of CNT forests with different CO<sub>2</sub> concentrations was also performed at growth temperatures of 730 °C, 770 °C, and 790 °C and on patterned catalysts. Though the growth behavior differed slightly at different growth temperatures, the same trend of morphology evolution from convex-shaped forests to bowl-like forests with increasing CO<sub>2</sub> concentration was observed (Figs. S-1–S-3 in the Electronic Supplementary Material (ESM)). The forests grown on the patterned catalyst also exhibited similar behavior in each lattice (Fig. S-4 in the ESM).

Nonuniform growths of CNT forests on the edge and in the middle of substrates have been widely reported [24, 34–37]. The structure of the forests will be discussed in the following section to illustrate the nonuniform self organization of CNTs.

## 2.2 Modulating the structure of CNTs in the forests

The as-grown CNT forests were further characterized by high-resolution electron microscopy. The top surface structure, cross-section view, and fine structure of CNTs obtained under various CO<sub>2</sub> concentrations are illustrated in Fig. 2. It is commonly believed that during the formation of CNT forests, the CNTs firstly construct a woven structure by entangling with each other. With continuous growth of

the CNTs, a vertically aligned CNTs forest structure can subsequently be self-assembled *via* synchronous growth [32]. From the left column in Fig. 2, it can be seen that an entangled CNT woven structure was formed, and the aligned CNTs are shown in the middle column of Fig. 2. Without the addition of CO<sub>2</sub>, the woven structure in the top of the forest was covered by a large quantity of carbon impurities such that single CNTs can hardly be distinguished in the SEM images (Fig. 2(a)). The carbonaceous impurities accumulated on CNT forests, especially on the top surface of forests, have been attributed to the pyrolysis of ethylene in the gas phase and deposition on the surface of the CNTs, and the accumulation becomes severe with prolonged CVD duration [38–40]. Figure 2(d) shows the surface obtained with a



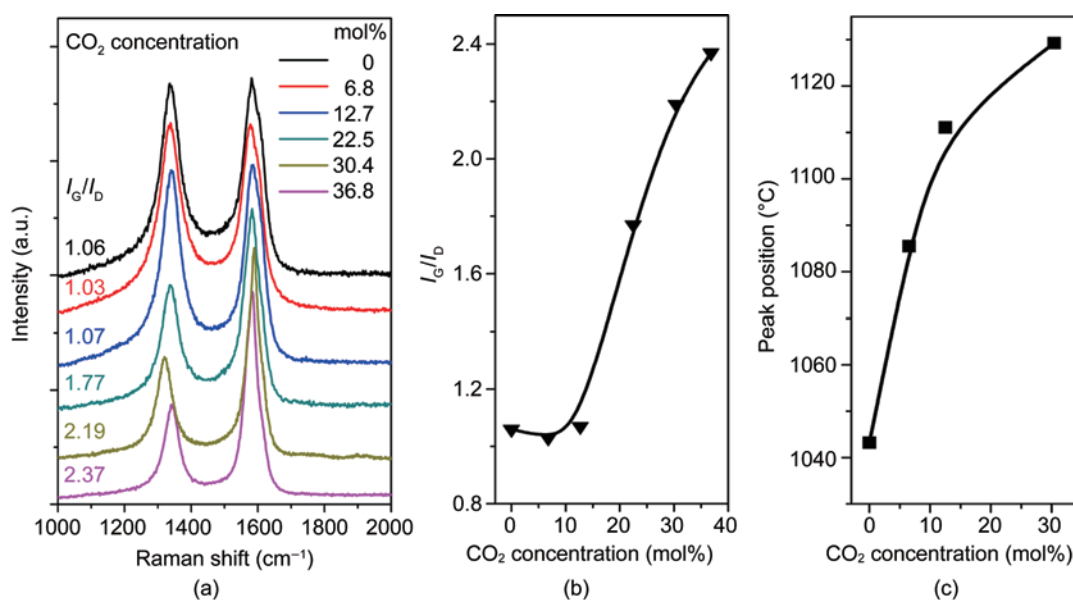
**Figure 2** Top surface morphologies of CNT forests (left column), side view of aligned CNT forests (middle column), and high-resolution TEM images of constituent CNTs (right column) of CNT forests obtained with CO<sub>2</sub> concentrations of 0 mol% (top row), 12.5 mol% (middle row), and 30.4 mol% (bottom row)

low CO<sub>2</sub> concentration (12.5 mol%). The top woven structure was appreciably cleaner than that obtained without CO<sub>2</sub> addition, although there were still some carbonaceous impurities coating the CNTs. When further increasing the concentration of CO<sub>2</sub> (to 30.4 mol%), amorphous carbon was mostly eliminated, leaving clean and bare CNTs on the top surface of the CNT forests (Fig. 2(g)). Meanwhile, these CNTs were found to be sparser and less entangled compared with the CNT forests obtained with lower or no CO<sub>2</sub> addition.

As described above, the amount of carbonaceous impurities varied with the concentration of CO<sub>2</sub> in the CVD process. Raman spectroscopy was employed to identify the carbonaceous impurities. Spots were taken in the middle of the side of the CNT forests obtained with various CO<sub>2</sub> concentrations (Fig. 3(a)). In a typical Raman spectrum, the G band (near 1580 cm<sup>-1</sup>) and the D band (near 1340 cm<sup>-1</sup>) represent the resonance peak of graphite and the scattering peak of the disordered component, respectively. Therefore, the intensity ratio ( $I_G/I_D$ ) is commonly used to characterize the crystallization degree of CNT samples. The variation in  $I_G/I_D$  with CO<sub>2</sub> concentration is shown in Fig. 3(b). The  $I_G/I_D$  was maintained at about 1.0 until the CO<sub>2</sub> concentration was over 15 mol%, after which the  $I_G/I_D$  increased dramatically and almost linearly.

The  $I_G/I_D$  for the CNT forest obtained with a CO<sub>2</sub> concentration of 36.8 mol% reached 2.37, indicating a relatively low content of carbonaceous impurities. This plot confirms that the CO<sub>2</sub> did not significantly oxidize the sample at low CO<sub>2</sub> concentrations and appreciable oxidation only occurred when the concentration was over 15 mol%, after which the oxidation effect increased with the increasing CO<sub>2</sub> concentration. To compare the stability of CNT forests obtained under different CO<sub>2</sub> concentrations, TG analysis under a CO<sub>2</sub> atmosphere was employed. The temperature of the mass loss peak appeared to increase with increasing CO<sub>2</sub> concentration during the synthesis process, indicating the enhanced stability of CNTs synthesized in the presence of CO<sub>2</sub> arising from the selective oxidation (Fig. 3(c)).

The high-resolution TEM images (right column in Fig. 2) show the fine structure of the as-grown CNTs in the forest under different conditions, and reveal the differences in the diameter and the wall number. Briefly, the forest obtained without CO<sub>2</sub> mainly consisted of double-walled CNTs (DWCNTs) (Fig. 2(c)) while single-walled CNTs are in the majority when higher CO<sub>2</sub> concentrations were employed (Fig. 2(i)). The wall number and diameter distributions of CNTs in forests obtained under different conditions were calculated based on the



**Figure 3** (a) Raman spectra of CNT forests obtained with addition of different amounts of CO<sub>2</sub>, (b) plot of the  $I_G/I_D$  ratio of the CNT forests, and (c) the peak positions of the mass loss (TG under CO<sub>2</sub> atmosphere) for CNT forests obtained with different CO<sub>2</sub> concentrations

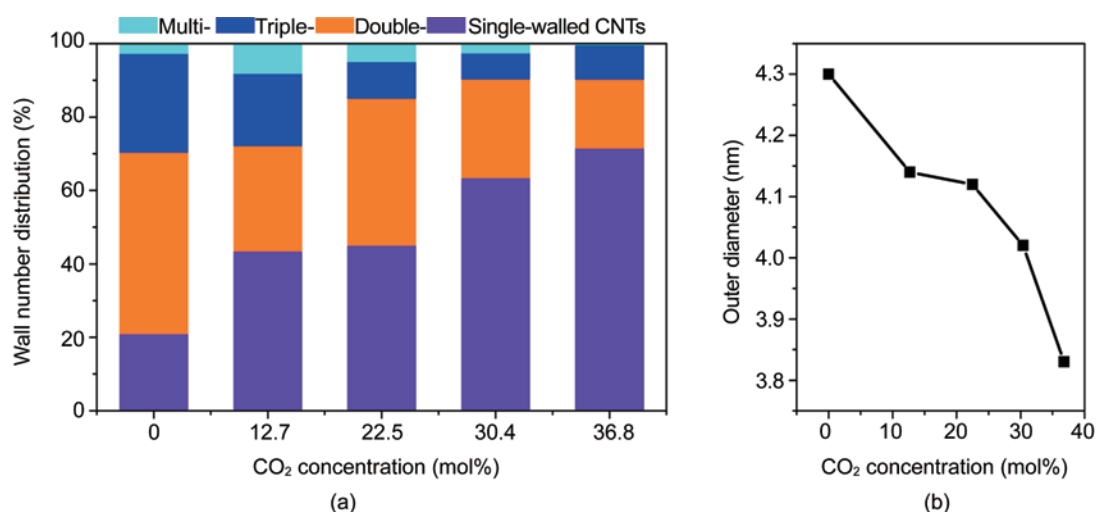
statistics for over two hundred CNTs. Figure 4(a) illustrates the wall number distribution of the as-grown CNTs. As indicated by the stack bars, without CO<sub>2</sub> addition the content of double-walled CNTs was near 50% and the content of single-walled CNTs was below 20%. For CNTs grown with the assistance of CO<sub>2</sub>, the content of single-walled CNTs increased gradually with increasing concentration of CO<sub>2</sub>. The content of single-walled CNTs was near 40% when the CO<sub>2</sub> concentration was 12.7 mol% and 22.5 mol%. When the CO<sub>2</sub> concentration was further increased to 30.4 mol% and 36.8 mol%, the content of single-walled CNTs reached 63% and 71%, respectively. Together with the increase in content of single-walled CNTs, formation of CNTs with large wall numbers (triple- and multi-walled) was not observed at high levels of CO<sub>2</sub>. Besides its role in modulating the wall number distribution, the presence of CO<sub>2</sub> in the growth atmosphere can also change the diameter of CNTs. As indicated by Fig. 4(b), the mean diameters of CNTs in the forest were negatively correlated with the concentration of CO<sub>2</sub>. By introducing CO<sub>2</sub> with a concentration of 36.8 mol%, the average outer diameter decreased from 4.30 nm to 3.83 nm. The mean diameters of the inner shell of CNTs were around 3.8 nm and did not differ significantly with changing CO<sub>2</sub> concentration, indicating that the addition of CO<sub>2</sub> did not obviously alter the size of the iron particles.

Since the fine structure of CNTs in the forest can be

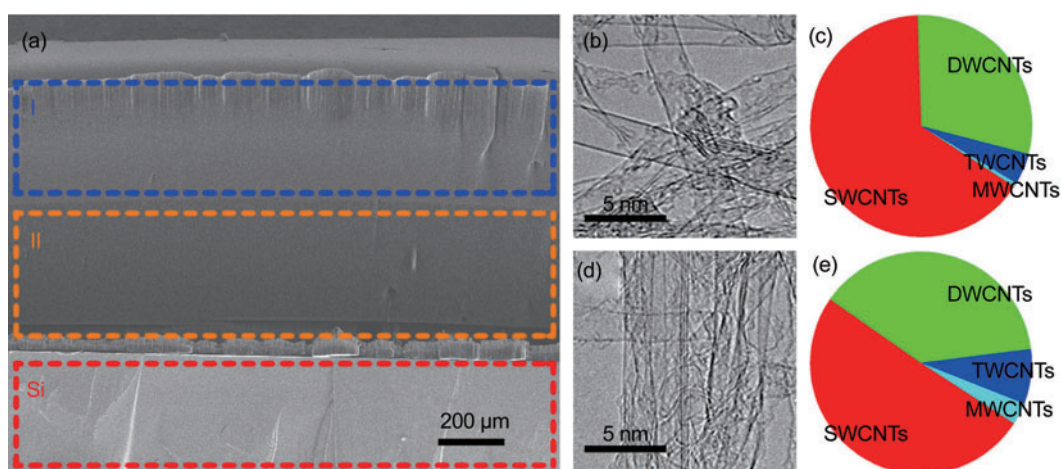
modulated by the concentration of CO<sub>2</sub> introduced, a two-stage growth of CNTs was conducted to further demonstrate the effect of CO<sub>2</sub> on the growth of CNTs. The two CO<sub>2</sub> concentrations were 30.4 mol% and 6.5 mol%, and the duration of each stage was 30 min. According to the SEM image shown in Fig. 5(a), the forest can be divided into two layers and there was an obvious interface at the middle section of the forest. The heights of the two layers were 380 μm and 320 μm, respectively. The forest possessed a similar bowl-like shape to those obtained in the presence of CO<sub>2</sub> described above. The first layer (Part I) of the forest was grown under higher CO<sub>2</sub> concentration, leading to a relatively high content of single-walled CNTs, as indicated by the high-resolution TEM image and the pie chart (Figs. 5(b) and 5(c)). The content of the single-walled CNTs was 65.3% while the content of double-walled CNTs was 29.5% and CNTs with large wall number were almost entirely absent. Due to the relatively low concentration of CO<sub>2</sub> during the growth of the bottom (Part II) of the forest, the CNTs in this part were composed of more graphite layers. The contents of single-walled CNTs and double-walled CNTs were 50.5% and 38.3%, respectively, leaving over 10% of triple- or multi-walled CNTs in the forest.

### 2.3 Mechanism of the tuning effect induced by CO<sub>2</sub>

As shown in the above section, both the forest morphologies and CNT structures were modulated



**Figure 4** (a) Changes in wall number distribution and (b) mean outer diameter of CNTs in forests obtained with different CO<sub>2</sub> concentrations



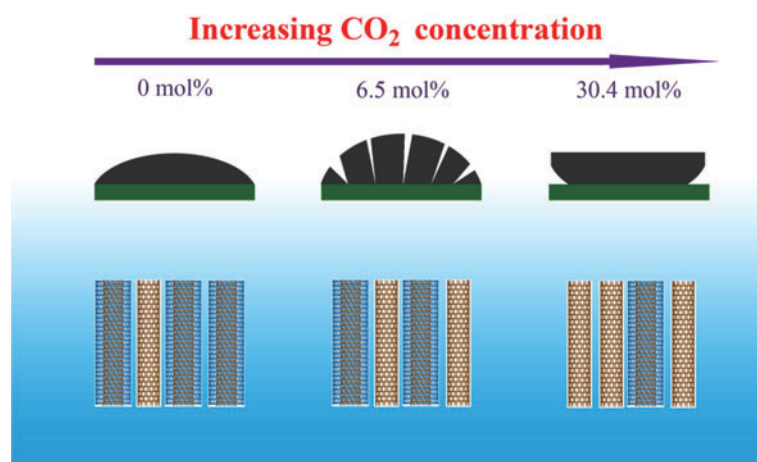
**Figure 5** (a) SEM image of two-layered CNT forest, with I indicating the first layer and II representing the second layer; high-resolution TEM images of constituent CNTs of (b) I and (d) II; pie charts of the wall number distribution of CNTs in (c) I and (e) II

by  $\text{CO}_2$  addition. In the CVD process, the introduced  $\text{CO}_2$  can react with  $\text{H}_2$  and generate  $\text{H}_2\text{O}$ , which may boost the growth of CNT forests. Excess  $\text{CO}_2$  also acts as a weak oxidant which reacts with the deposited solid carbon, resulting in the observed changes in the forest morphologies and the structure of the CNTs (Fig. 6).

The changes in the morphologies of the CNT forests can be mainly attributed to the changes in the growth rate of the CNT forests and the modulation of the CNT-substrate adhesive force. Millions of CNTs synchronously grow into a forest [32]. In our case, the density of CNTs in the array was estimated to be  $2300\text{--}2600\ \mu\text{m}^{-2}$ , which is lower than “super growth” ( $5200\ \mu\text{m}^{-2}$ ) [24], alcohol assisted growth ( $8500\ \mu\text{m}^{-2}$ ) [41], or plasma enhanced growth ( $14000\ \mu\text{m}^{-2}$ ) [14]. The sparse density of CNTs suggests that the ability to form the initial top woven structure is a key factor. In the middle section of the substrate, CNTs can easily form the woven structure and grow into a high forest [32], while at the edge, the difficulty in forming the woven structure limits the growth rate, and a CNT convexity is formed. The nonuniformity in growth rate also induces internal stress distributed throughout the CNT forests [32, 37]. With the growth of CNT forests, the stress accumulates. Due to the strong bonding between CNT forests and the substrate, the

forests invariably split on the top surface to release the internal stress. Without  $\text{CO}_2$  addition, only a few cracks form due to the difference in growth rates on the edge and in the middle of the substrate, and the whole forest possesses a convex shape. With the addition of  $6.5\ \text{mol}\% \text{CO}_2$ , the growth of CNT forests in the middle part is dramatically boosted (from  $0.4\ \text{mm/h}$  without  $\text{CO}_2$  to  $1.1\ \text{mm/h}$  under  $6.5\ \text{mol}\% \text{CO}_2$ ), which enlarges the nonuniformity in growth rate. Accordingly, the CNT forests split into a large number of blocks and many cracks form.

For high  $\text{CO}_2$  concentrations, the bowl-like forests were formed due to the reduced CNT-substrate adhesive force. Generally, CNTs bond firmly to the substrate with the catalyst between them. The



**Figure 6** Schematic diagram of the evolution of the forest morphology (middle panel) and the approximate resulting wall number distribution of CNTs in the forest (bottom panel) with increasing  $\text{CO}_2$  concentration

presence of certain substance can etch the catalyst or the deposited carbon near the catalyst and reduce the CNT-substrate adhesive force [42]. The CNTs near the iron particles were preferentially etched due to the catalytic oxidation by high concentrations of CO<sub>2</sub>. At the initial stage of CNT growth, the forest split on the edge because of the weak bonding between CNT forests and the substrate, leaving some residual forest on the edge of substrate. With the further growth of the forest, the outermost part of the forest continuously detached from the substrate, resulting in the bowl-shaped morphology. It is worth mentioning that the stress in the forests was released by the rupture of forests from the substrates and the nonuniformities of growth rate were eliminated, leaving a complete top surface structure of the forests.

The CO<sub>2</sub> also changes the structure of CNTs in the forest through oxidative etching. Small amounts of CO<sub>2</sub> reacted with H<sub>2</sub> and the oxidative etching of solid carbon was not obvious. However, higher concentrations of CO<sub>2</sub> can cause the oxidation of both as-grown CNTs and carbonaceous impurities. Generally, the amorphous carbon and the incomplete graphite layers react more readily with CO<sub>2</sub> compared with the well-graphitized CNT walls. Thus, the amorphous carbon contained in the CNT forests could be completely eliminated by increasing CO<sub>2</sub> addition and the  $I_G/I_D$  increased accordingly (Fig. 3(b)). Accompanying the etching of amorphous carbon, the complete graphite layers of CNTs (especially the outer walls of CNTs) were inevitably slightly etched by CO<sub>2</sub>. As a result, this process can tune the wall number distribution of the CNTs in the forests. The wall number of the CNTs decreased gradually with increasing CO<sub>2</sub> concentration.

Finally, it should be noted that by CO<sub>2</sub> addition, the morphology and CNT selectivity in the as-grown forest can be modulated easily. With the increase of CO<sub>2</sub> concentration from 0 mol% to 36.8 mol%, the content of single-walled CNTs increased from 20% to over 70%. Even for CNT forests radially grown on spheres [33], flying carpets on flakes [43], forests intercalated in lamellar clays [44] and the odako growth of dense CNT forests [45], CO<sub>2</sub> addition can be expected to play an important role in easily

modulating the growth behavior both in a fixed bed or a fluidized bed reactor.

### 3. Conclusions

Introduction of CO<sub>2</sub> into the CVD process allows controllable synthesis of single- or double-walled CNT forests. Both the CNT forest morphology and CNT structure were modulated by the introduction of CO<sub>2</sub>. When the CO<sub>2</sub> concentration was increased from 0 mol% to 30.4 mol%, the morphology of CNT forests changed from convex-shaped to radial-block-shaped and then to bowl-shaped. By introducing large amounts of CO<sub>2</sub>, the carbonaceous impurities can be eliminated and the wall number of CNTs can also be significantly reduced. When the CVD atmosphere comprised 36.8 mol% of CO<sub>2</sub>, the content of single-walled CNTs in the forest reached 70%. It is suggested that the addition of CO<sub>2</sub> generates water and serves as a weak oxidant to modulate both the morphology of the forest and the structure of the CNTs. CO<sub>2</sub>-assisted growth is therefore a facile way for the controllable synthesis of CNT forests, and also provides a further general understanding of process intensification for nanostructure modulation.

### Acknowledgements

The work was supported by the National Natural Science Foundation of China (Nos. 20736007 and 2007AA03Z346), and the China National Basic Research Program (No. 2006CB0N0702).

**Electronic Supplementary Material:** SEM images of CNT forests grown with varying concentrations of CO<sub>2</sub> at different temperatures on a silica substrate as well as on a patterned catalyst are available in the online version of this article at <http://dx.doi.org/10.1007/s12274-009-9088-6> and are accessible free of charge.

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